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Short Communication

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Youn-Geun Kim, Manuel P. Soriaga

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CATHODIC REGENERATION OF A CLEAN AND ORDERED Cu(100)-(1×1) SURFACE
FROM AN AIR-OXIDIZED AND DISORDERED ELECTRODE:
AN *OPERANDO* STM STUDY

YOUN-GEUN KIM¹ AND MANUEL P. SORIAGA¹

Joint Center for Artificial Photosynthesis
California Institute of Technology
Division of Chemistry and Chemical Engineering
Pasadena, CA 91125

¹Corresponding authors. Email Addresses: msoriaga@caltech.edu. ygkim@caltech.edu.

ABSTRACT

In work related to the electrocatalysis of the CO₂ reduction reactions, we recently reported in *This Journal* the structure and composition of a Cu(100) electrode surface, pre-dosed at low levels of O_{2(g)} to simulate a Cu electrocatalyst unprotected from air, before and after immersion in alkaline electrolyte at fairly negative potentials to ascertain if an oxide-to-metal reduction reaction can be effected; experimental measurements were based upon *ex situ* techniques, low-energy electron diffraction (LEED) and Auger electron spectroscopy (AES). It was found that the mildly oxidized surface remained ordered and could be cathodically reduced back to a well-ordered oxide-free Cu(100); the quality of the LEED pattern and AES spectrum was less than ideal, however, due to small amounts of base electrolyte remnant in the emerged layer. In this *Short Communication*, we present results from *operando* electrochemical scanning tunneling microscopy (EC-STM) that not only confirm the earlier observations but, more importantly, depict more accurately the actual electrocatalysis conditions. An as-received commercially oriented Cu(100) disk that had not been protected from air was observed to consist of narrow terraces encrusted with highly disordered oxides. Cyclic voltammetry and coulometry showed that the oxidized surface consisted of five monolayers of CuO and quarter of a monolayer of Cu₂O. Upon complete cathodic reduction of the interfacial oxides, the surface was found to have reverted to a single-crystalline Cu(100)-(1×1) structure. It may thus be inferred that, under the conditions of electrochemical CO₂ reduction, the Cu catalyst would exist as a zerovalent metal.

KEYWORDS

Operando Electrochemical Scanning Tunneling Microscopy

Copper Electrocatalysts for CO₂ Reduction

Cathodic Regeneration of Ordered Cu(*hkl*) from Oxidized Surfaces

Air-oxidized Copper Electrocatalyst Surfaces

In Situ Electrochemical Scanning Tunneling Microscopy

INTRODUCTION

The electrochemical reduction of CO₂ catalyzed by Cu is strongly influenced by the crystallographic orientation of the catalyst surface [1-3]: Cu(111) preferentially forms CH₄, Cu(100) favors C₂H₄, and Cu(pc) may yield up to sixteen different products. However, unless the electrodes are prepared and maintained under a stringently oxygen-free environment, the copper surface would always contain several layers of oxide, CuO and/or Cu₂O, since the material is notorious for its high affinity towards gaseous oxygen [4,5]. It is of fundamental importance to understand how pure-copper electrocatalysis is enabled given the initial presence of an interfacial oxide crust.

In an earlier study published in *This Journal* [6], clean and highly ordered Cu(*hkl*) single-crystal surfaces were exposed to low levels of O_{2(g)}, nominally fifteen minutes at 10⁻⁶ Torr, as a controlled means to simulate a Cu electrocatalyst unprotected from air [5]. The surfaces, which were only mildly oxidized and displayed ordered structures [e.g., Cu(100)-(√2×√2)R45°-O and Cu(110)-(2×1)-O], were then immersed in alkaline solutions at fairly negative potentials. *Ex situ* surface interpellation by low-energy electron diffraction (LEED) and Auger electron spectroscopy (AES) after emersion indicated that the surfaces had been reduced back to the well-ordered, oxide-free metallic states; the quality of the AES spectra and LEED patterns, however, was less than ideal due to small amounts of base electrolyte remnant in the emersed layer.

In this *Short Communication*, we describe results from *operando* electrochemical scanning tunneling microscopy (EC-STM) of an as-received, commercially prepared Cu(100) single-crystal electrode that had been exposed to atmospheric oxygen for an extended period. The present work is a more accurate representation of the first-time use of actual copper electrocatalysts.

EXPERIMENTAL

A crystallographically oriented Cu(100) disk, 99.999 % in purity, 1.0 mm in thickness, and 10 mm in diameter, was purchased commercially (Princeton Scientific Corp, Princeton, NJ). The crystal, which had been freely handled in ambient-atmosphere conditions for an unspecified period, was purposely exposed to air for another seven days. It was then used for the EC-STM experiments, in deaerated 0.10 *M* KOH (Sigma-Aldrich, St. Louis, MO), without pretreatment other than sonication in acetone followed by copious rinses with MilliQ water (Millipore Corporation, Billerica, MA).

EC-STM was carried out with an Agilent 5500 microscope (Agilent Technologies, Santa Clara, CA) equipped with a three-electrode potentiostat. An electrochemical cell was custom-crafted from Kel-F® (Emco Industrial Plastics, Inc.), fitted with a Pt counter and a Pt pseudo-reference electrode calibrated against an SHE reference cell. The STM tips were prepared by an electrochemical etch of a 0.25 mm diameter tungsten wire (Sigma-Aldrich) in 1.0 *M* KOH at *ca.* 15 V. All images were obtained with a high-resolution scanner in a constant-current mode without post-scan processes such as with high-pass filters.

RESULTS AND DISCUSSION

Figure 1 shows a low-resolution (500 nm x 500 nm) *in situ* STM image of the air-exposed Cu(100) electrode surface at the rest or open-circuit potential (OCP), *ca.* 0.1 V vs. SHE, in 0.1 M KOH. The existence of only narrow (high step-density) and topographically rounded terraces is easily discernible; the excessive step density, which is atypical of single-crystal surfaces, is due to the fact that thermal or electrochemical pretreatments had been carefully avoided. The terraces are completely covered with a highly disordered surface-oxide crust that could not be imaged at higher resolution. *Operando* EC-STM at potentials between 0.1 V and -0.1 V did not reveal any changes in the image displayed in Figure 1.

The chemical nature of the air-oxidized electrode surface was examined by cyclic voltammetry. The dashed curve in Figure 2 represents the first voltammetric cycle, initially from the OCP in the direction of more negative potentials. Two cathodic peaks are observed, a minor one at -0.32 V and a rather large, broad peak at -0.80 V. Neither of these two is present in subsequent (solid curve) scans. In the reverse, anodic-direction scan, only one peak appears at 0.03 V. This anodic peak is also found in later current-potential cycles, along with a cathodic peak at -0.38 V. It is clear from Figure 2 that the steady-state anodic and cathodic peaks represent, respectively, the forward and reverse processes of the same interfacial redox reaction; in conformity to published studies [7,8], the reaction is attributed to the kinetically irreversible $\text{Cu}_2\text{O}_{(s)}/\text{Cu}_{(s)}$ monolayer-limited surface redox process. It thus appears that the small first-scan (dashed curve) cathodic peak in Figure 2 is a low-coverage $\text{Cu}_2\text{O}_{(s)}$ -to- $2\text{Cu}_{(s)}$ reduction, with the shift in peak potential most likely due to its presence in the outer vicinity of the thick oxide crust.

It is evident in Figure 2, and in agreement with earlier work [7,8], that the fairly sizable cathodic peak centered at -0.8 V, can only be ascribed to the $\text{CuO}_{(s)}$ -to- $\text{Cu}_{(s)}$ reduction. In controlled experiments on the anodic oxidation of Cu at potentials more positive than required for $2\text{Cu}_{(s)}$ -to- $\text{Cu}_2\text{O}_{(s)}$ oxidation, it was found that the morphology of the $\text{CuO}_{(s)}$ -to- $\text{Cu}_{(s)}$ reduction

peak was influenced by the thickness of the higher oxide film. However, the breadth of the $\text{CuO}_{(s)}\text{-to-Cu}_{(s)}$ peak in Figure 2, stretched from -0.5 V to -1.0 V, is exceptional; other interfacial factors, undocumented at this time, are almost certainly in play.

The thickness of the surface oxide film, in terms of the number of monolayers of CuO and Cu_2O , was obtained from the Faradaic charges for the *first-scan* reduction peaks referenced against that for the *steady-state* Cu_2O reduction; because the latter is surface-limited, by virtue of Faraday's Law, it defines the charge for one monolayer. The experimental data indicate 5 monolayers of CuO and 0.25 monolayer of Cu_2O .

Figure 3 shows a low-resolution (500 nm x 500 nm) *operando* EC STM image at -0.9 V after cathodic reduction of the CuO and Cu_2O crust. It is most noteworthy that, compared to the image of the oxidized electrode in Figure 1, the surface is now constituted of wider and flatter terraces. It is discernible that the density of step-lines has decreased; its persistence is due to the absence of thermal pretreatment prior to the experiments.

The high-resolution (5 nm x 5 nm), atom-resolved image of the oxide-free terraces is displayed in Figure 4. It reveals a square (1x1) lattice. While the terraces are not particularly wide, the atomic arrangement within a given terrace is highly ordered. The interatomic distance was measured to be 0.27 ± 0.01 nm, a value identical to that for a pristine, oxide-free square Cu(100) net. The regeneration of a clean and well-ordered Cu(100)-(1x1) surface by cathodic reduction of the surface oxide is thus established.

CONCLUSIONS

The cathodic regeneration of a clean and well-ordered Cu(100) electrode from a disordered air-oxidized surface, has been studied by *operando* electrochemical scanning tunneling microscopy. It was found that a commercially oriented Cu(100) sample, previously exposed to ambient air for over a week, consisted of narrow (high step-density) terraces encrusted with highly disordered oxides. Cyclic voltammetry and coulometry showed that the surface oxides were made up of five monolayers of CuO and quarter of a monolayer of Cu₂O. Upon complete cathodic reduction of the oxide film, the surface was observed to have reverted to a single-crystalline Cu(100)-(1×1) structure. It may thus be inferred that, under conditions typical of electrochemical CO₂ reduction, the copper electrocatalyst would exist as a zerovalent metal.

ACKNOWLEDGMENTS

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FIGURES

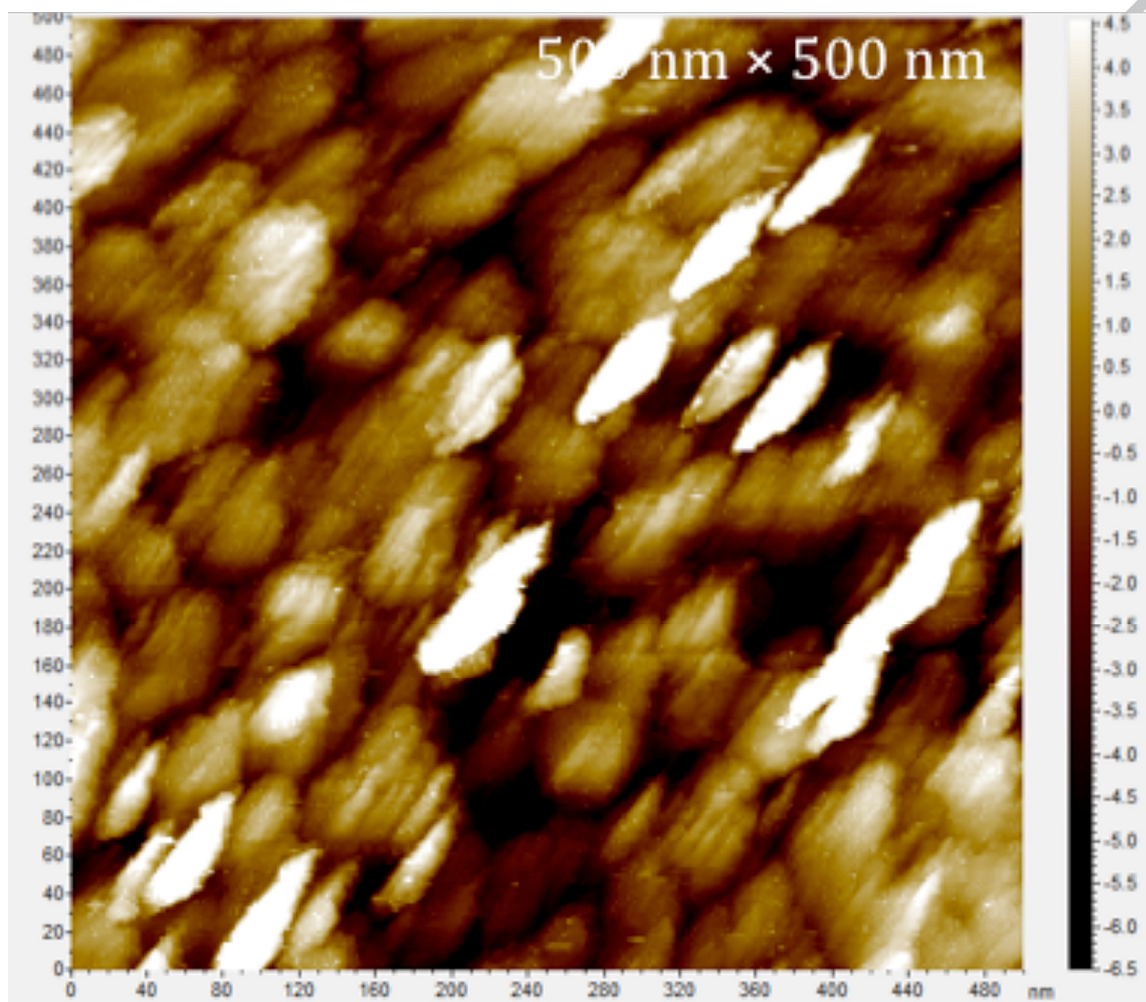


Figure 1. Low-resolution *in situ* STM image of the air-oxidized Cu(100) electrode surface at the open-circuit potential in 0.1 M KOH. Experimental parameters: Bias voltage = 250 mV; tunneling current, 2 nA.

Voltammetry of Oxidized and Cathodically Regenerated Cu(100)

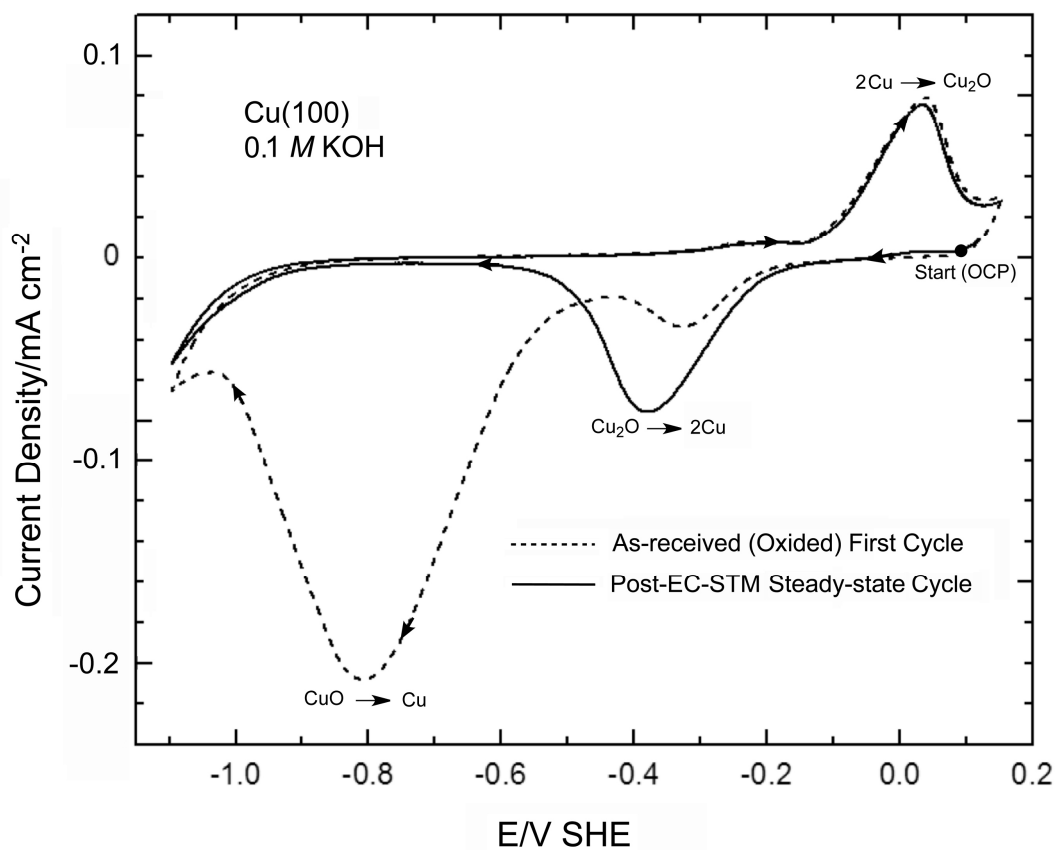


Figure 2. Cyclic voltammetric curves for the oxidized (dashed curve: first scan) and cathodically regenerated (solid curve: steady-state scan) Cu(100) in 0.1 M KOH. The potential scan rate was 50 mV sec^{-1} ; the geometric area of the electrode was 1.84 cm^2 .

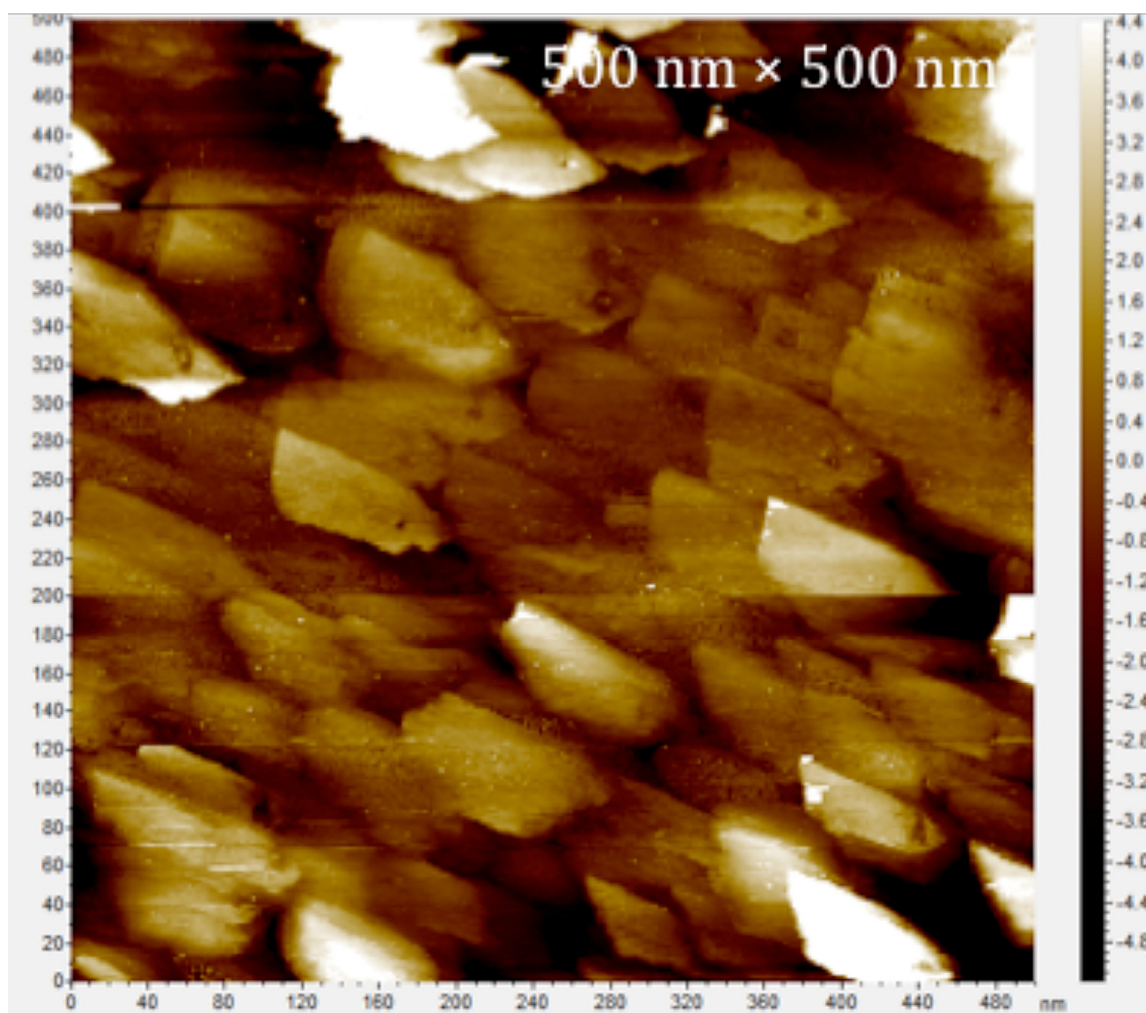


Figure 3. Low-resolution *operando* STM image, at -0.9 V in 0.1 M KOH, of the cathodically regenerated Cu(100) electrode surface. Experimental parameters: Bias voltage = -120 mV; tunneling current, 2 nA.

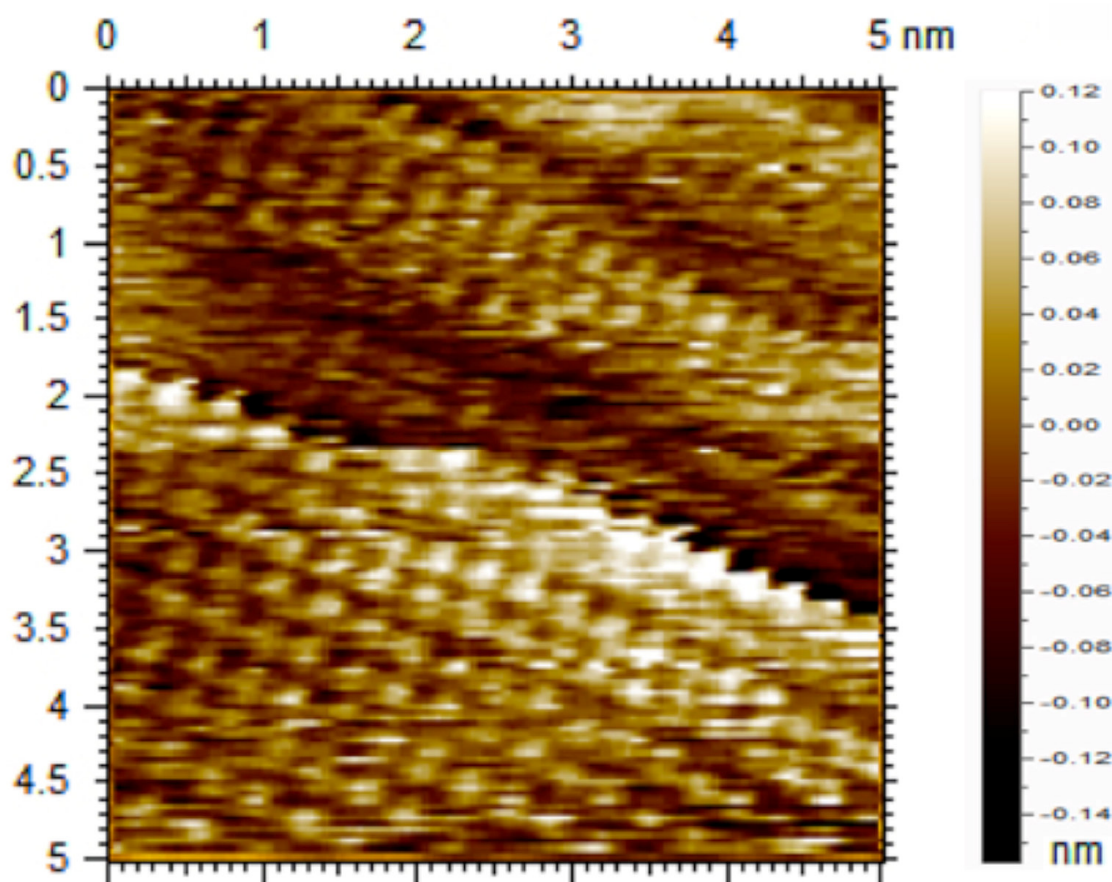


Figure 4. High-resolution *operando* STM image, at -0.9 V in 0.1 M KOH, of the cathodically regenerated Cu(100) electrode surface. Experimental parameters: Bias voltage = -120 mV; tunneling current, 5 nA.

Fig. 1

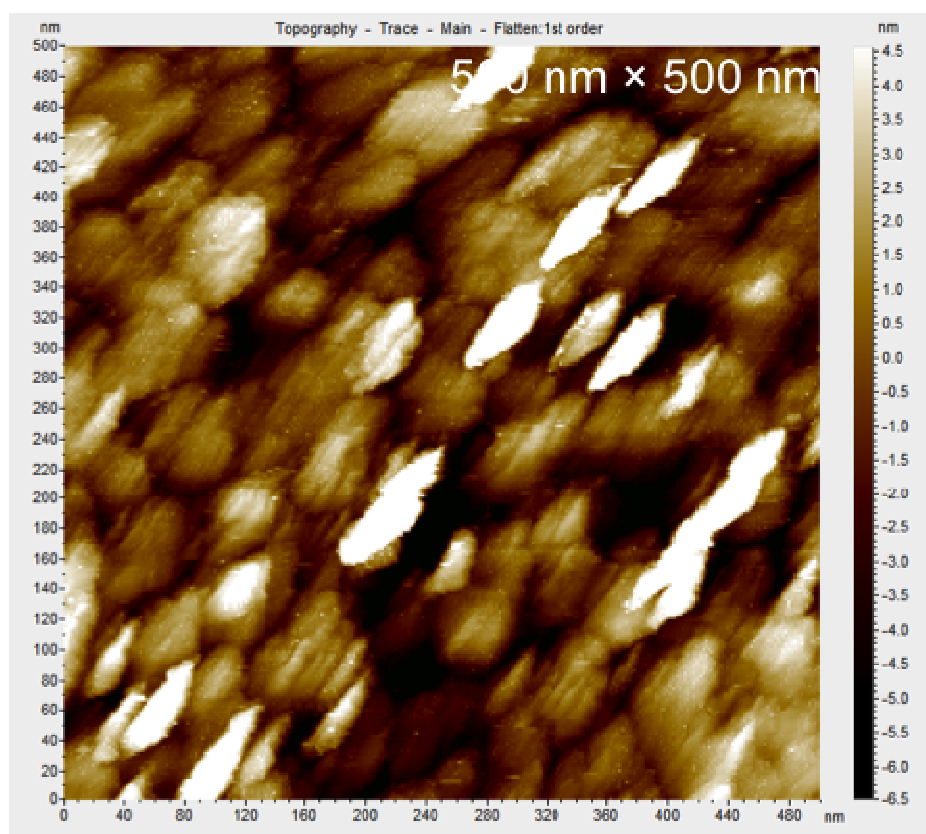
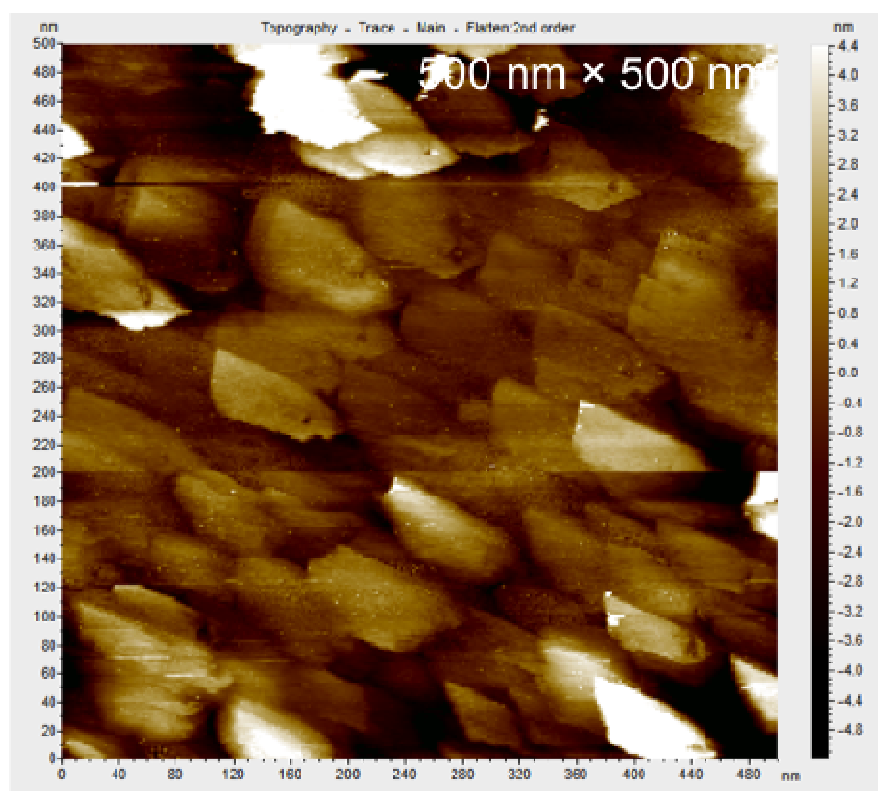


Fig. 3



Highlights

- The cathodic regeneration of a clean and well-ordered Cu(100) surface from an electrode pre-exposed to air was studied by *operando* electrochemical scanning tunneling microscopy.
- The multilayer oxide crust on the Cu(100) surface exposed to atmospheric oxygen was highly disordered.
- The air-oxidized Cu(100) surface consisted predominantly of CuO with minor amounts of Cu₂O.
- The multilayer oxide crust on the Cu(100) surface was completely reduced back to the metal after only one cathodic voltammetric scan.
- Upon completion of the cathodic reduction of the interfacial oxides, the surface reverted to a clean and well-ordered Cu(100)-(1×1) structure.